

Physicochemical and tribological properties of gemini-type halogen-free dicationic ionic liquids

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Abstract: A series of new halogen-free dicationic ionic liquids (ILs) with different alkyl chain lengths were prepared, and the relationship between the alkyl chain length, physicochemical and tribological properties of ILs, and their role as neat lubricant for steel–steel friction pairs, was investigated. Evaluation of stability during hydrolysis and copper strip corrosion test results show that synthetic ILs are stable and not corrosive to metal contacts, due to the halogen-free anions. The friction and wear test results indicate that ILs with long alkyl chains have excellent friction-reducing and anti-wear properties, especially at high temperatures. Based on the surface three-dimensional (3D) profiles, electrical contact resistance, scanning electron microscopy (SEM), energy dispersive X-ray spectroscopy (EDS), and the X-ray photoelectron spectrometry (XPS) analysis of the worn surfaces of steel discs, we can conclude that the efficiency of ILs is due to the formation of high quality tribofilms that consist of both tribochemical reaction and ordered absorption films.

Keywords: halogen-free; dicationic ionic liquids (ILs); lubricant; tribofilm

1 Introduction

Friction and wear result in energy wastage and shorten the service life of mechanical components [1]. The reasonable use of liquid lubricants is one of the most effective ways to prevent friction and wear. In general, liquid lubricants are mainly divided into two categories—natural and synthetic lubricants. Natural lubricants include some animal fat, vegetable oils, and mineral oils [2]. Synthetic lubricants commonly contain synthetic hydrocarbon, synthetic ester, and perfluoropolyethers (PFPEs), and are widely used in special conditions, such as applications that require high temperature stability, low saturated vapor

pressures, excellent extreme pressure, and oxidation stability [3].

The discovery of ionic liquids (ILs) as high performance synthetic lubricants was initiated in 2001 [4]. The study of ILs as lubricant has since received increasing attention [5–8]. ILs have been explored as lubricants for various frictional materials, due to their excellent dipolar structure and physical properties like extremely low volatility, nonflammability, high thermal stability, low melting point, and good thermal conductivity; the latter allowing for rapid frictional heat dissipation [9–14]. Currently, a large number of studies in the tribological field are mainly focused on halogen-containing mono-cation ILs based on anions

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such as $[\text{BF}_4]^-$, $[\text{PF}_6]^-$, $[\text{CF}_3\text{SO}_3]^-$, and $[\text{NTf}_2]^-$ [15–17]. In 2006, dicationic ionic liquid lubricants were first reported by Ref. [18]. In this study, polyethylene glycol functionalized dicationic ILs with alkyl or polyfluoroalkyl substituents were prepared. It is noteworthy that these ILs also exhibit excellent tribological characteristics even at 300 °C. A new series of dicationic symmetrical and asymmetrical ILs consisting of tetraalkylphosphonium and alkylimidazolium were synthesized, which have very high decomposition temperature of approximately 450 °C and good tribological properties [19]. Pagano et al. [20] synthesized and investigated twelve dicationic ILs as potential lubricant, which were synthesized from oligoethylene glycols linking two cationic moieties based on either N-methylimidazolium or N-methylpyrrolidinium at the extremities. As anions, chloride, bis(trifluoromethanesulfonyl)imide $[\text{NTf}_2]^-$, methanesulfonate and butanesulfonate were chosen. Mahrova et al. [21] synthesized a series of bi-cationic pyridine-type ILs, and systematically studied their tribological properties as lubricant and additive. This series of ILs uses $[\text{NTf}_2]^-$ as an anion and exhibits excellent friction-reduction and anti-wear properties. However, ILs containing $[\text{BF}_4]^-$, $[\text{PF}_6]^-$, $[\text{CF}_3\text{SO}_3]^-$, and $[\text{NTf}_2]^-$ usually undergo hydrolysis when used as lubricants. They can cause metal corrosion and can pollute the environment [22]. Hence, replacement of ILs with halogen-free high-efficiency IL lubricants is becoming increasingly popular in modern ILs tribology [23–28]. Gusain et al. [29] synthesized and evaluated a series of bis(imidazolium)- and bis(ammonium)-di[bis(salicylato)borate] ILs with variable alkyl chain and cyclic ring structures as potential lubricant additives. These dicationic ILs as additives of PEG200 showed noncorrosive properties, which were attributed to the of corrosion driven elements, such as halogens, in these ILs.

In this study, we designed a new type of halogen-free dicationic IL, based on the gemini-type molecular structure. The cation part is a double quaternary ammonium structure, and the anion is a phosphate ester. We adopted the N/P synergistic lubrication concept. The constructed ILs are excellent lubricants and have excellent thermal stability.

2 Experimental

2.1 Chemicals

The reagents, including N,N,N',N'-tetramethylethylenediamine, bromotetradecane, bromohexadecane, bromooctadecyl, bis(2-ethylhexyl) hydrogen phosphate, and sodium hydroxide were obtained from Sigma-Aldrich. Di-isooctyl phosphate sodium was prepared according to previously reported methods [30]. The molecular structures of 1,2-bis-N,N-dimethyl-N-tetradecylammonium bis (2-ethylhexyl) phosphate (NP-14-2-14), 1,2-bis-N,N-dimethyl-N-cetylammmonium bis (2-ethylhexyl) phosphate (NP-16-2-16), 1,2-bis-N,N-dimethyl-N-octadecylammonium bis (2-ethylhexyl) phosphate (NP-18-2-18), are shown in Fig. 1. The method of these three gemini-type ILs has been reported in detail by our previously reported Ref. [31]. The structures and purities of dicationic ILs (NP-14-2-14, NP-16-2-16 and NP-18-2-18) were finely confirmed by ^1H NMR, ^{13}C NMR, ^{31}P NMR, and MS spectroscopic data. Detailed data are presented in the Electronic Supplementary Material (ESM).

2.2 Thermal stability and viscosity analysis

Thermal stability and viscosity-temperature performance are the most basic characteristics of good lubricants. Therefore, thermogravimetric analysis (TGA)

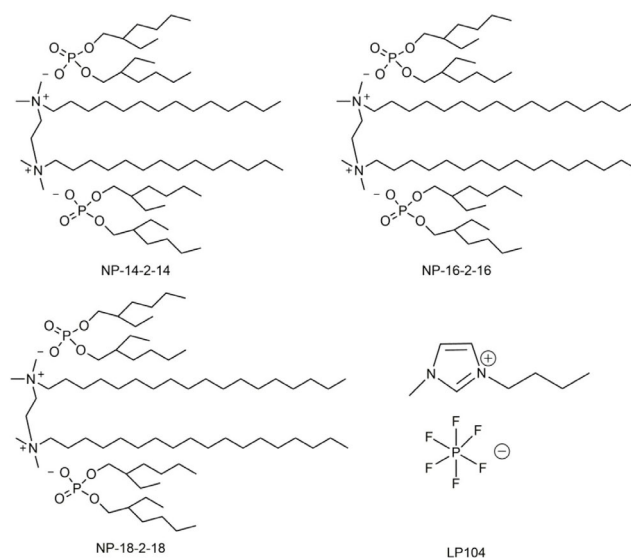


Fig. 1 Molecular structures of selected dicationic ILs and 1-butyl-3-methylimidazolium hexafluorophosphate (L-P104) as a comparative synthetic lubricant.

was conducted on a Netzsch STA 449 F3 synchronous thermal analyzer system, in a nitrogen atmosphere at a temperature of 10 °C/min, to test the thermal stability of PAO 10, L-P104, NP-14-2-14, NP-16-2-16, and NP-18-2-18. Kinematic viscosities of all samples were measured at temperatures of 40 and 100 °C using a SYP1003-III kinematic viscosity tester.

2.3 Copper strip corrosion test

The copper strip corrosion test was performed according to the National standard GB-T5096-1985(91) [32]. Prior to the test, copper strips with a certain specification (12.0 mm × 12.0 mm × 3.0 mm) were polished to brightness and completely cleaned with ethanol. The copper strips were then immersed into the test lubricants (PAO 10, L-P104, and NP-18-2-18). The samples were heated at 100 °C for 24 h. After the experiment, the copper strips were taken out and washed carefully with ethanol, and then the corrosion level was gauged using a standard corrosion plate for comparison.

2.4 Hydrolysis stability

The hydrolytic stability experiments of NP-18-2-18 refer to previously reported methods [33]. NP-18-2-18 was mixed with water in equal quantity. The solutions were then heated at 80 °C for 24 h. The pH of the solutions was checked at an interval of 30 min to determine the hydrolysis levels of NP-18-2-18. In comparison, the hydrolysis stability of PAO 10 and L-P104 was measured simultaneously using the same method.

2.5 Tribology test and surface analysis

The tribological properties of NP-14-2-14, NP-16-2-16, and NP-18-2-18 for the steel–steel friction pair were evaluated using a reciprocating ball-on-disk sliding test on an Optimol SRV-V. Table 1 shows the test parameters based on previously reported methods [34, 35], at environmental humidity of 30%–50%, using an upper slide ball (model specifications: AISI 52100 bearing steel; diameter: 10 mm; hardness: 59–63 HRC; mean roughness: 100 nm) and lower stationary steel disk (ø 24 mm × 7.9 mm, hardness: 59–63 HRC). A Bruker NPFLEX non-contact surface mapping profiler was used to determine the profile of the wear scar

Table 1 Test parameters of the SRV test.

Parameter	Value
Friction pairs	Steel/steel
Load (N)	300
Frequency (Hz)	25
Temperature (°C)	25/100
Amplitude (mm)	1
Duration (min)	30
Voltage (V)	20

and the corresponding wear volume.

An FEI Quanta FEG 250 SEM with EDS (type: Oxford IE250) was used to observe the morphology and composition of the surface of wear scars. XPS (type: PHI-5702) was used to further observe the binding energy of elements distributed on the surface of the wear scar, to better understand the lubrication mechanism.

3 Results and discussion

3.1 Physicochemical properties

3.1.1 Thermal analysis

The thermal stability of all lubricants was tested, and the results are presented in Fig. 2 and Table 2. It can be observed that the thermal decomposition temperatures (T_d) of PAO 10 (~280.5 °C) and L-P104 (~400 °C) are very high, whereas the T_d values of NP-14-2-14 (~212.4 °C), NP-16-2-16 (~233.1 °C), and NP-18-2-18 (~238.1 °C) are relatively low. The thermal decomposition temperatures of dicationic IL were also found

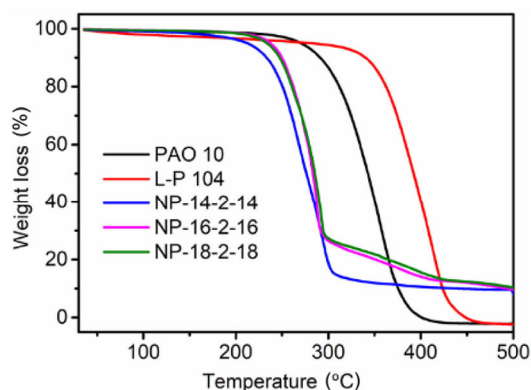


Fig. 2 TGA curves of PAO 10, L-P104, NP-14-2-14, NP-16-2-16, and NP-18-2-18.



Table 2 Kinematic viscosities (KV) at 40 °C (KV₄₀), 100 °C (KV₁₀₀), and VIs of dicationic ILs and the reference samples.

Lubricant	KV _s		VI	T_d (°C)	TG temperature (°C) per weight loss		
	KV ₄₀ (mm ² /s)	KV ₁₀₀ (mm ² /s)			10%	20%	50%
PAO 10	63.75	9.64	133.10	280.47	290.40	310.40	341.90
L-P104	83.80	10.97	117.60	380.97	383.53	404.73	443.23
NP-14-2-14	2,856.10	108.51	107.10	212.44	233.10	250.30	276.30
NP-16-2-16	3,365.10	125.89	113.40	233.07	249.70	261.90	283.30
NP-18-2-18	3,529.90	156.85	137.30	238.77	247.20	260.40	285.50

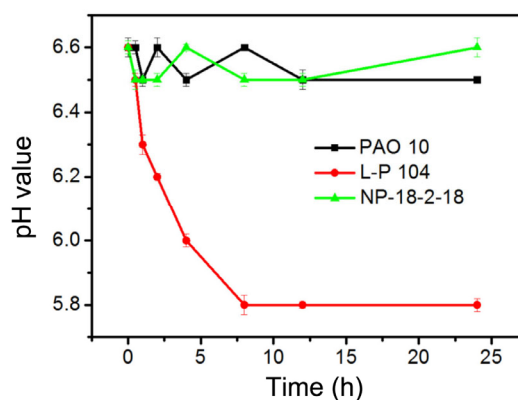
to increase slightly with increasing length of the alkyl chain, which can be attributed to an increase in Van der Waals interactions among the IL's molecules [18]. Table 2 shows the corresponding T_d for different mass losses. T_d shows an obvious gradual increase with a loss of weight. For example, T_d of PAO 10 indicates a 50% weight loss at ~ 340 °C, and the T_d of L-P104 indicates the same at 440 °C. There is 50% weight loss for all dicationic ILs at a temperature exceeding 250 °C. This suggests that synthetic ILs have appropriate thermal stability and can meet the lubrication requirements both at room temperature (RT) and high temperature (HT).

3.1.2 Viscosity

KVs and viscosity indexes (VIs) are also summarized in Table 2. The viscosities of dicationic ILs are obviously higher than those of PAO 10 and L-P104 at RT and 100 °C. The viscosity index of NP-14-2-14, NP-16-2-16, and L-P104 is approximately 110; PAO 10 and NP-18-2-18 have the highest viscosity indexes at approximately 130. The KVs and VIs of the dicationic IL lubricants gradually increase with the increase in alkyl chain length. This is attributed to the increase in Van der Waals interactions between the IL's molecules [36].

3.1.3 Hydrolysis stability

Compared to conventional halogen-containing ILs, halogen-free ILs demonstrate excellent hydrolytic stability [15]. In this study, the hydrolysis stability of halogen-free dicationic ILs was first evaluated by using a hydrolysis test according to Ref. [22]. Figure 3 shows no significant pH change during the test, which indicates that PAO 10 and halogen-free dicationic ILs are hydrolysis stable. Under the same test conditions,

**Fig. 3** pH value change curves of the samples.

L-P104 showed rapid hydrolysis with a drop in pH value at the very beginning of the test.

3.1.4 Copper strip corrosion test

Traditional halogen-containing ILs can attack substrates due to their hydrolysis, to produce acidic materials [37, 38]. The purpose of designing halogen-free dicationic ILs is to improve the lubricating properties of ILs and reduce their corrosion performance. Therefore, the copper strip corrosion test was developed to confirm anti-corrosion properties, using PAO 10 and L-P104 as references. The photographs of the polished copper strips (Fig. 4(a)) after soaking in PAO 10 (Fig. 4(b)), L-P104 (Fig. 4(c)), and NP-18-2-18 (Fig. 4(d)), respectively, are shown in Fig. 4. Compared to blank copper, it is evident that the surface of the copper sheet immersed in L-P104 is found to be corroded. This surface looks tarnished and is covered with a large amount of corrosion products (Fig. 4(c)), while only a small amount of darkening was observed on the surfaces of the copper strips soaked in PAO 10 (Fig. 4(b)) and NP-18-2-18 (Fig. 4(d)). According to the standard corrosion plate, it can be concluded that

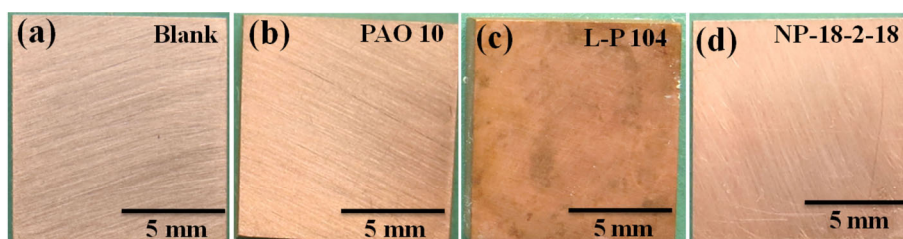


Fig. 4 Photographs of copper strips (a) before and (b–d) after the corrosion test: (b) PAO 10, (c) L-P104, and (d) NP-18-2-18.

the corrosion grade for L-P104 was 2a, whereas those for PAO 10 and NP-18-2-18 were 1a.

3.2 Analysis of friction and wear properties as lubricants for steel–steel friction pairs

The friction coefficient (COF) and wear volume of NP-14-2-14, NP-16-2-16, and NP-18-2-18 were first tested at RT (Fig. 5). Figure 5(a) shows that the COF of PAO 10 is very high at 0.6 and then gradually reduces to an equilibrium of 0.2, which is attributed to the non-polar property of PAO molecules in that they do not easily adsorb on the metal interface. Furthermore, there are no special elements such as N and P in the molecule that can cause frictional chemical reactions; thus, PAO 10 has a very high wear volume of up to $1.2 \times 10^{-3} \text{ mm}^3$ (Fig. 5(b)). The COFs of NP-14-2-14, NP-16-2-16, and NP-18-2-18 are low and stable. A significant decrease in COF is observed in the entire process, with an increase in alkyl chain length. This law is consistent with the experimental results of wear volume (Fig. 5(b)), suggesting that the alkyl chain length of ILs has an important effect on the stability of the boundary lubricant film, considering the lubrication interval. The average COF for all synthetic ILs is less than 0.1, which is a good numerical

range for a good boundary lubrication area according to Ref. [39]. Compared to PAO 10, NP-14-2-14, NP-16-2-16, and NP-18-2-18, L-P104 exhibited a very low and smooth COF and relatively low wear volume. It is easy to understand that L-P104 is more suitable as a lubricant for steel–steel system at RT, mainly because of the contribution of active element content (high F, P, and N element percentage) in molecules and viscosity.

Under 100 °C, the COF of PAO 10 has equal shares at RT (Fig. 6(a)), but indicates a greater wear volume of approximately $2.0 \times 10^{-3} \text{ mm}^3$ during the sliding process (Fig. 6(b)). The final COF of NP-18-2-18 is less than that of L-P104. The anti-wear properties of NP-16-2-16 and NP-18-2-18 are also superior to those of L-P104 (Fig. 6(b)). As expected, the reduction of the COF and wear volume of NP-16-2-16 and NP-18-2-18 is evident, indicating that the friction reduction and anti-wear performance of dicationic ILs can be enhanced by introducing long alkyl chains at HT. Furthermore, a suitably high lubricant viscosity is beneficial for forming a thicker lubricating film, since long alkyl chains are superior for stronger hydrophobic interactions at HT. The formation of a stable lubricating film is thus caused by a

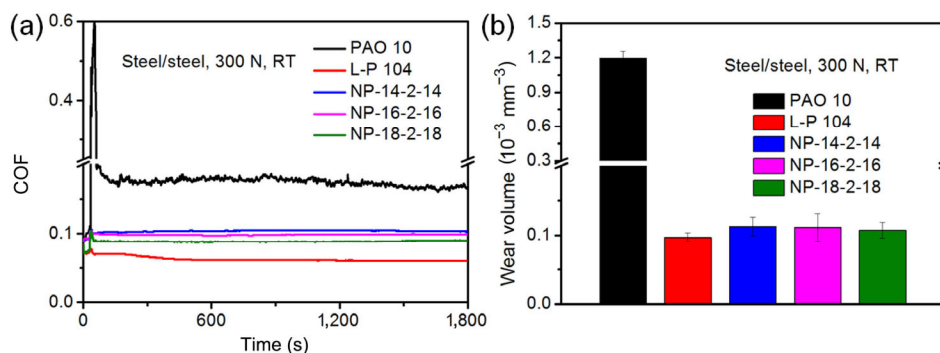


Fig. 5 Evolution of (a) COF/time and (b) wear volumes of PAO 10, L-P104, NP-14-2-14, NP-16-2-16, and NP-18-2-18 for steel–steel contact at 300 N, 25 Hz, and RT.



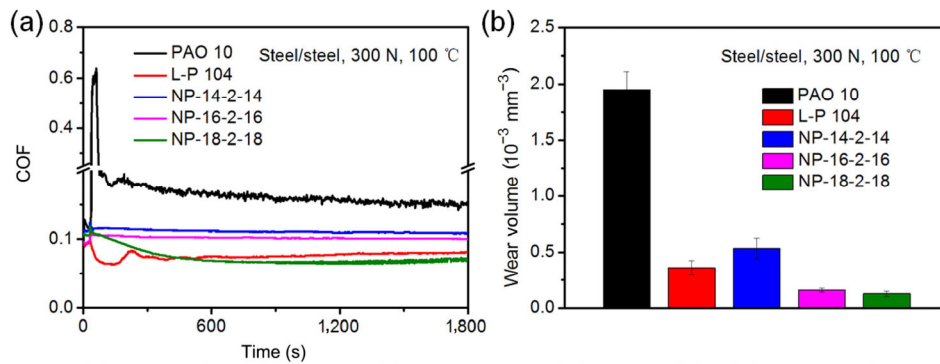


Fig. 6 Evolution of (a) COF/time and (b) wear volumes of PAO 10, L-P104, NP-14-2-14, NP-16-2-16, and NP-18-2-18 for steel–steel contact at 300 N, 25 Hz, and 100 °C.

strengthened interaction and enhanced lubrication in NP-16-2-16 and NP-18-2-18.

3.3 Surface analysis of wear spot

3.3.1 Surface profile analysis

The three-dimensional (3D) optical microscopic images of the wear spots at RT and 100 °C, are exhibited in Figs. 7 and 8, respectively. The results are consistent with the wear volume analysis. The wear scar diameter (WSD) and wear scar depth (WSD') of PAO 10 are 2.3 mm and $-6 \mu\text{m}$, respectively (Fig. 7), and there are obvious protrusions around the wear scar. In comparison, the WSD and WSD' values of other lubricants

were significantly reduced. For example, for a lubricated surface of L-P104 (Fig. 7(b)), the WSD was approximately 1.2 mm, and the WSD' was approximately $-0.5 \mu\text{m}$. The same phenomenon was observed for other dicationic ILs lubricants (Figs. 7(c)–7(e)). However, if we look closely at Figs. 7(c)–7(e), the reduction in WSD and WSD' is obvious. These results also prove that as the alkyl chain increases, the lubricating properties of the ions also increase.

At 100 °C, the 3D morphology of the wear surface is similar to that at RT. The obvious difference is that the wear spot of the PAO 10 lubricant has a larger WSD and a deeper WSD' (Fig. 8(a)). For surfaces lubricated by L-P104 (Fig. 8(b)) and NP-14-2-14

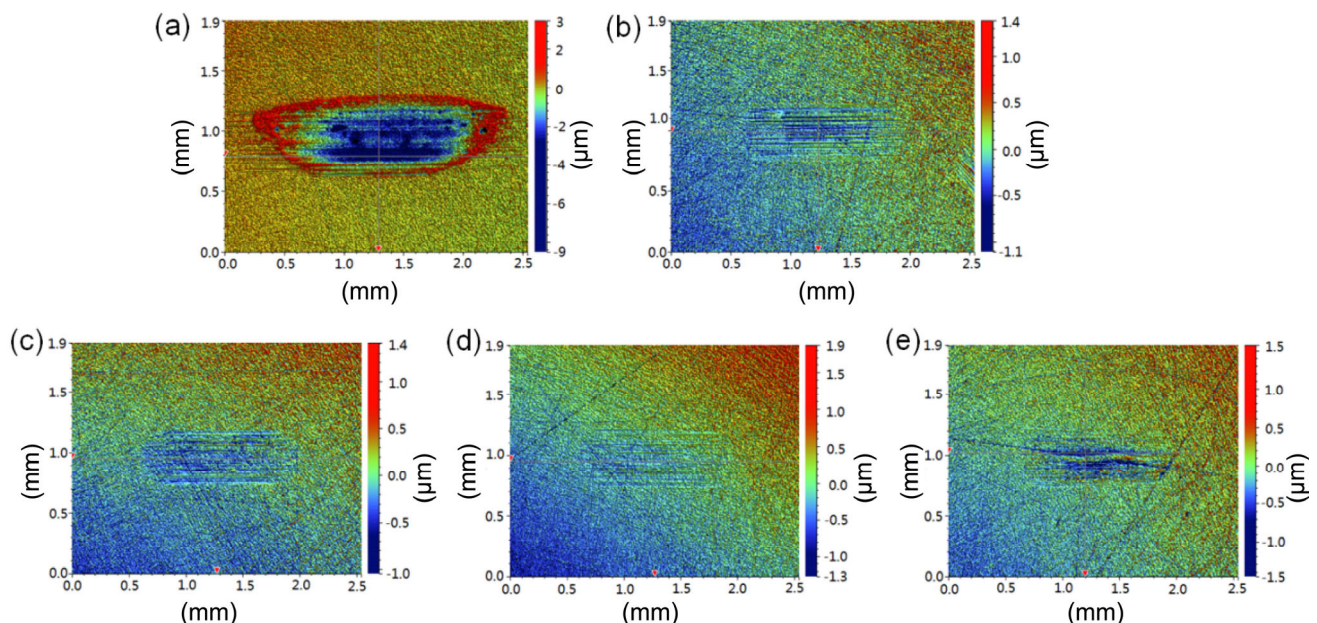


Fig. 7 3D optical microscopic images of wear tracks (steel–steel friction pairs) corresponding to (a) PAO 10, (b) L-P104, (c) NP-14-2-14, (d) NP-16-2-16, and (e) NP-18-2-18 at RT.

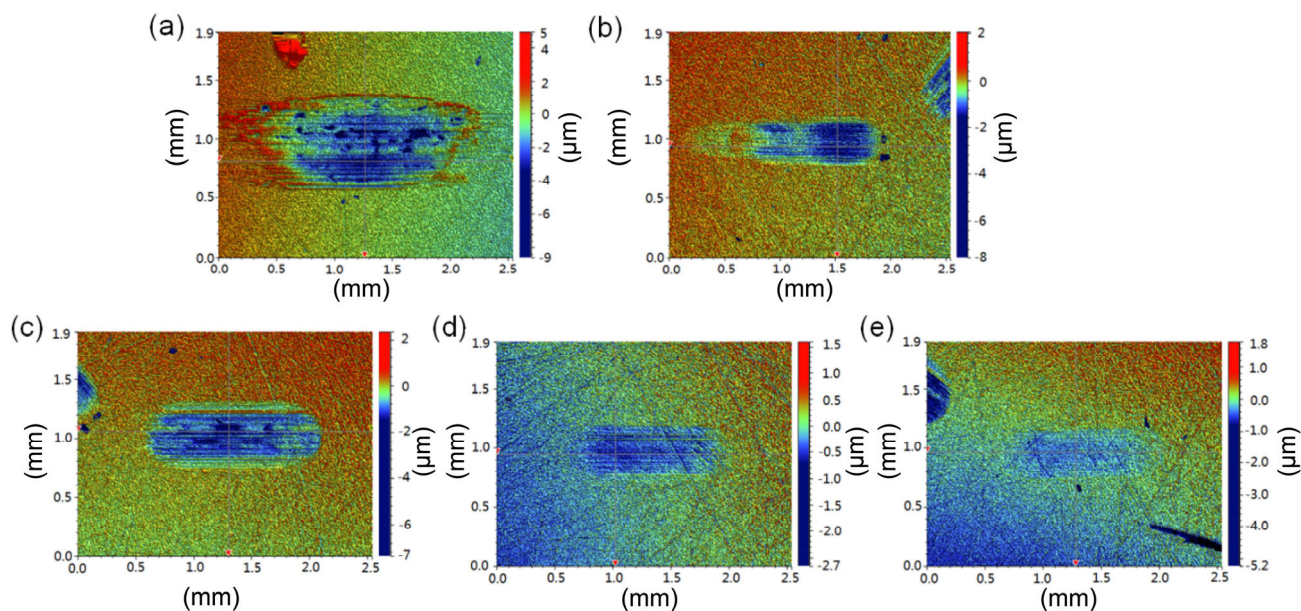


Fig. 8 3D optical microscopic images of wear tracks (steel–steel friction pairs) corresponding to (a) PAO 10, (b) L-P104, (c) NP-14-2-14, (d) NP-16-2-16, and (e) NP-18-2-18 at 100 °C.

(Fig. 8(c)), the wear was significantly heavier than that at RT. However, for other lubricants, the increase in WSD and WSD' is small (Fig. 8(b)–8(e)) at 100 °C compared to at RT. It is worth noting that as the alkyl chain length increases, WSD and WSD' decrease significantly, which means that high-viscosity ILs are more advantageous for reducing friction and wear at HTs.

3.3.2 SEM analysis

The SEM images show worn surfaces of the lubricated steel at RT and 100 °C (Figs. S1 and S2 (ESM)). Using ILs as lubricants at RT significantly reduces wear. For example, from Figs. S1(a)–S1(e) (ESM), it can be seen that the wear spot diameter is significantly reduced, and the surfaces of wear become better. However, SEM topographies magnified to 1,000 times indicate that obvious differences exist between various lubricants. For example, steel surfaces lubricated with PAO 10 (Fig. S1(a1) (ESM)) and L-P104 (Fig. S1(a2) (ESM)) have obvious adhesive wear, as observed under an SEM at a magnification of 3,000 times. Deep pits appear on the surface that was lubricated with PAO 10 (Fig. S1(a2) (ESM)), suggesting a failure of lubrication. The lubricated surface of L-P104 (Fig. S1(a2) (ESM)), with a large black area indicates metal surface corrosion caused by anion hydrolysis. When observed clearly, the worn surfaces lubricated by NP-14-2-14,

NP-16-2-16, and NP-18-2-18 have neither obvious scratches nor corrosion, implying excellent anti-wear and anti-corrosion properties of IL NP-14-2-14, NP-16-2-16, and NP-18-2-18.

3.4 Simple analysis of lubrication mechanism

3.4.1 Electrical contact resistance (ECR) analysis

Using ECR, measurements can offer a visualization of the change in interface resistance and the formation of an interface tribofilm [40, 41]. During friction, ECR changes were gauged from the lubrication interface at RT (Fig. 9(a)) or at HT (Fig. 9(b)), and the results are consistent with those of the wear volume analysis. Combined with Figs. 5, 6, and 9, it can be easily concluded that a reduction in friction is related to the formation of tribofilm on worn surfaces, which is caused by changes in the ECR values. For example, the ECR of PAO 10 is very low at about 0.1 Ω regardless of whether it is at RT or 100 °C, and this corresponds to the highest COF. In contrast, the ECR values of dicationic IL lubricants changed from 0.1 to higher ECR values and a lower COF was observed. More importantly, the ECR increases significantly with an increase in alkyl chain length for dicationic ILs. This change is more pronounced at HT, suggesting that ILs are more likely to form an insulating lubricating



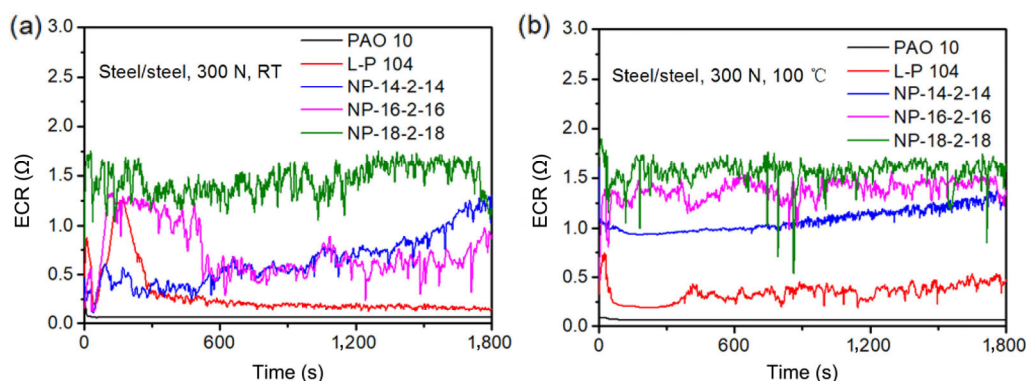


Fig. 9 ECR of PAO 10, L-P104, NP-14-2-14, NP-16-2-16, and NP-18-2-18 as lubricants for steel–steel at (a) RT and (b) 100 °C with a load of 300 N and frequency of 25 Hz.

protective film at the interface through adsorption or tribochemical reaction during the rubbing process, thus providing better anti-friction and anti-wear properties. In contrast, although L-P104 exhibits low friction at both RT and HT, its ECR is not as high as that of the dicationic ILs. We suspect that the change in ECR is closely related to the conductivity of the lubricant. To further explain this phenomenon, the conductivity of L-P104, NP-14-2-14, NP-16-2-16, and NP-18-2-18 were tested, and shown in Table S1 of the supporting information. As can be seen from Table S1 (ESM), the conductivities of L-P104, NP-14-2-14, NP-16-2-16, and NP-18-2-18 were 1,451, 7.31, 6.98, and 6.33 $\mu\text{S}/\text{cm}$, respectively. Compared with L-P104, dicationic ILs have a very low conductivity, which results in a high contact resistance during the rubbing process. In addition, compared with L-P104, dicationic ILs have a longer alkyl chain, and form a denser and thicker lubricating film [42], and thus their ECR values are larger.

3.4.2 EDS analysis

The friction test at RT showed the chemical composition of wear scars, which was confirmed by EDS. Figures S3(a)–S3(e) (ESM) represent elements in the wear spot after lubrication by different lubricants, respectively. In Figs. S3(a) and S3(a1)–S3(e1) (ESM), elements C, O, and Fe appear on the surface of the wear scar and on the outside of the wear scar, suggesting that these lubricants do not react with the metal during friction. Figures S3(a)–S3(e) (ESM) show some special elements, such as F, P, and N that appear on the center of the wear scar, indicating that NP-14-2-14,

NP-16-2-16, and NP-18-2-18 are consistent with L-P104, which undergo a complex tribochemical reaction with the steel, forming a stable boundary lubrication film. This reduces friction and enhances anti-wear properties [43].

With friction at HT, the element distribution on the surface of the wear scar is exactly the same as the element distribution after friction at RT.

3.4.3 XPS analysis

To gain further information about the lubricating effect of the halogen-free dicationic ILs, XPS analysis of the boundary film on worn steel surfaces following friction was conducted for elements such as Fe, O, P, and N. Figure 10 shows the XPS curve of the worn surface lubricated with NP-16-2-16 and L-P104, at RT and 100 °C. Figure 10 indicates that the XPS spectra of surfaces lubricated with NP-16-2-16 and L-P104 are quite similar, both at RT and HT. In the case of the Fe 2p XPS spectrum (Fig. 10(a)), Fe 2p peaks appear at approximately 710.8 and 724.5 eV, associated with the binding energy (BE) of 530.2 and 531.8 eV in the O 1s spectrum (Fig. 10(b)). They can be identified as Fe_2O_3 , Fe_3O_4 , and/or $\text{Fe}(\text{OH})\text{O}$ [44]. As shown in Fig. 10(c), the BE of P 2p at about 133.8 eV might correspond to compounds that contain PO_4^{3-} [45–47], and the P 2p at 134.80 eV is attributed to alkyl phosphate. In addition, the peak of P 2p at 133.8 eV is assigned to FePO_4 , which coincides with the Fe 2p peak of 712.80 eV. This indicates that NP-16-2-16 decomposes and reacts with the exposed metal surface to produce a protective film composed of phosphates and/or polyphosphates, which improves the tribological

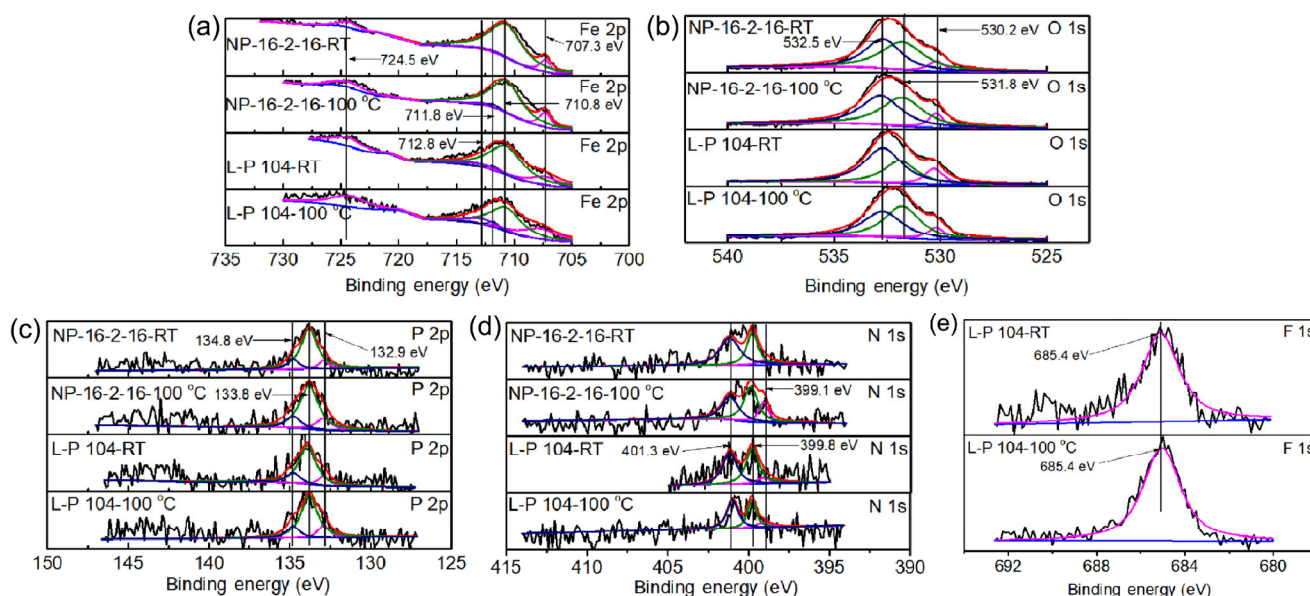


Fig. 10 XPS spectra of (a) Fe 2p, (b) O 1s, (c) P 2p, (d) N 1s, and (e) F 1s lubricated by NP-16-2-16 and L-P104 at RT and HT (load = 300 N; stroke = 1 mm; frequency = 25 Hz; duration = 30 min).

properties [47]. From Fig. 10(d), the N 1s peak of 401.30 eV indicates a complex phosphorus oxide species, namely $(\text{NH}_4)_2\text{HPO}_4$, and the O 1s peak of 531.50 eV also indicates this compound [48]. The N 1s peak of 399.80 eV may correspond to C–N bonding and nitrogen being transformed into organic amines [49, 50]. The N 1s peak at 399.10 eV is attributed to phosphinic amide, which coincides with the P 2p peak of 132.90 eV. The above results together indicate that a new nitride and phosphorus compound such as FePO_4 and complex organic nitrogen compounds were generated on the worn surfaces during the sliding process [47]. For a surface lubricated by L-P104, as shown in Fig. 10(e), the XPS peaks of F 1s, P 2p, and Fe 2p appear at 685.4, 133.8, and 711.8 eV, respectively, which are assigned to FeF_2 and FePO_4 [46].

4 Conclusions

A series of halogen-free dicationic ILs equipped with long alkyl chains were synthesized and evaluated as lubricants for a steel–steel friction pair at RT and HT, with the aim of studying the relationship between the alkyl-chain length and the tribological properties of double cationic ILs. Increasing the alkyl chain length of the cation has proven to be effective in improving ILs in terms of thermal stability, corrosion

resistance, and tribological performance, especially at HT. Based on the results of COF, SEM, ECR, EDS, and XPS, the excellent lubrication of ILs comes from the boundary lubricating protection film containing phosphorus and nitrogen that forms on the sliding steel surfaces, which achieves superior tribological properties by a phosphorus–nitrogen synergy.

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